Imaging oxygen defects and their motion at a manganite surface

BRYANT, B., et al.

Abstract

Manganites are technologically important materials, used widely as solid oxide fuel cell cathodes; they have also been shown to exhibit electroresistance. Oxygen bulk diffusion and surface exchange processes are critical for catalytic action, and numerous studies of manganites have linked electroresistance to electrochemical oxygen migration. Direct imaging of individual oxygen defects is needed to underpin understanding of these important processes. Currently, it is not possible to collect the required images in bulk, but scanning tunnelling microscopy (STM) could provide such data for surfaces. Here, we report the first atomic resolution images of oxygen defects at a manganite surface. Our experiments also reveal defect dynamics, including oxygen adatom migration, vacancy-adatom recombination and adatom bistability. Beyond providing an experimental basis for testing models describing the microscopics of oxygen migration at transition-metal oxide interfaces, our work resolves the long-standing puzzle of why STM is more challenging for layered manganites than for cuprates.

Reference


DOI : 10.1038/ncomms1219
Manganites are technologically important materials, used widely as solid oxide fuel cell cathodes; they have also been shown to exhibit electroresistance. Oxygen bulk diffusion and surface exchange processes are critical for catalytic action, and numerous studies of manganites have linked electroresistance to electrochemical oxygen migration. Direct imaging of individual oxygen defects is needed to underpin understanding of these important processes. Currently, it is not possible to collect the required images in bulk, but scanning tunnelling microscopy (STM) could provide such data for surfaces. Here, we report the first atomic resolution images of oxygen defects at a manganite surface. Our experiments also reveal defect dynamics, including oxygen adatom migration, vacancy-adatom recombination and adatom bistability. Beyond providing an experimental basis for testing models describing the microscopics of oxygen migration at transition-metal oxide interfaces, our work resolves the long-standing puzzle of why STM is more challenging for layered manganites than for cuprates.
Mixed-valence manganites, the basic units of which are manganese-centred oxygen octahedra, are used extensively in solid oxide fuel cell cathodes\textsuperscript{4,6–8,12}. They have attracted considerable attention because of their colossal magnetoresistance\textsuperscript{4}, in which electrical resistance is a strong function of an external magnetic field. Numerous groups have also examined electroresistance in manganite-based devices, in which a low-field electrical resistance is switched by the imposition of a higher field\textsuperscript{14–19}. Electroresistance has the potential to revolutionize microelectronics through the introduction of new components such as resistive random access memory cells\textsuperscript{7} and memristors\textsuperscript{1,15}. Electroresistance has been attributed, by several groups, to the electrochemical migration of oxygen ions\textsuperscript{16–18}. Diffusion of oxygen through the bulk material and at the surface are also crucial processes for catalytic action in fuel cell cathodes\textsuperscript{1}. Direct imaging of oxygen defects and of surface oxygen diffusion would be a distinct advantage for the understanding of these processes; time-resolved scanning tunnelling microscopy (STM) could in principle provide such images.

A key parameter in manganite materials is the formal valence state of the manganese ions, which is tuned by varying the mix of divalent and trivalent cations. Electroresistance is strongest close to 50\% divalent ion doping\textsuperscript{4,6–8,12}, in which the formal Mn valence is a half integer. Similarly, catalytic activity in manganites is enhanced by mixed valency\textsuperscript{4}. Oxygen defects such as excess oxygen ions and vacancies, implicated in both catalysis and electroresistance, are also most probable at half doping. Therefore, to maximize our ability to image oxygen defects in a manganite using STM, we require a half-doped manganite, the surface of which can be prepared reliably. Unlike the three-dimensional pseudocubic manganites, layered manganites include planes along which the material can be cleaved, producing an atomically flat surface. Accordingly, we have chosen to study Pr$\text{Sr}_2$Mn$_2$O$_7$, a half-doped bilayered manganite built from double sheets (Fig. 1a) of pseudocubic (Pr$\text{Sr}$)MnO$_3$ alternating with sheets of rock-salt-like (Pr$\text{Sr}$)O.

STM studies of manganites are uncommon, particularly when compared with those of the closely related high-temperature superconducting copper oxides. Previous experiments have demonstrated clear atomic resolution\textsuperscript{16–19} for pseudocubic manganites, but atomically resolved STM studies of layered manganites are rare\textsuperscript{20}, despite the fact that the layered structure provides cleaving planes.

In this paper, we report atomic resolution STM imaging of the bilayered manganite Pr$\text{Sr}_2$Mn$_2$O$_7$. In addition to atomic resolution on the atomically flat surface, oxygen defects, in the form of both adatoms and vacancies, are observed at the Pr$\text{Sr}_2$Mn$_2$O$_7$ surface. Time-resolved imaging has been used to demonstrate oxygen adatom migration and vacancy-adatom recombination; voltage-dependent oxygen motion and adatom bistability are also observed. Our results bring the benefits of atomic resolution to the study of manganite fuel cell cathode materials and electroresistive devices.

Results

Atomic resolution STM imaging of Pr$\text{Sr}_2$Mn$_2$O$_7$. Pr$\text{Sr}_2$Mn$_2$O$_7$ is semiconducting, and is antiferromagnetic below $T_N = 125$ K (ref. 15). Figure 1b shows the $c$ axis resistivity and differential resistivity of a Pr$\text{Sr}_2$Mn$_2$O$_7$ single crystal, as used for STM experiments; an inflection is seen in the resistivity at around $T_N$. Figure 2a is an STM topograph of cleaved Pr$\text{Sr}_2$Mn$_2$O$_7$, together with the crystal structure. Terraces several hundred nanometres across are visible, separated by steps of height $1.01 \pm 0.02$ nm, corresponding to $c/2 = 0.996$ nm as measured by X-ray diffraction\textsuperscript{11}. Atomic resolution can be achieved right across these terraces for this poor electrical conductor, in contrast to previous studies on La$_2$Sr$_5$Mn$_3$O$_{15+\delta}$ (ref. 20), which is a much better metal and in which screening of atoms is more effective. Figure 2b shows a high-resolution STM topograph displaying a square atomic lattice with $a = 0.40 \pm 0.01$ nm. This is in reasonable agreement with the value from X-ray diffraction of $a = 0.385$ nm (ref. 15). Figures 2b and 3a reveal, in addition to the atomic lattice, surface inhomogeneities with $\pm 20$ pm corrugation. The latter depend on bias polarity (see Supplementary Fig. S1) and therefore have an electronic component, most likely due to charge inhomogeneities (mixed Mn valence and/or cation distribution). These features

Figure 1 | Structure and properties of Pr$\text{Sr}_2$Mn$_2$O$_7$. (a) Crystal structure of Pr$\text{Sr}_2$Mn$_2$O$_7$ showing bilayered structure. (b) $c$ axis resistivity $\rho$ and differential resistivity $\partial \rho/\partial T$ of Pr$\text{Sr}_2$Mn$_2$O$_7$ versus temperature. The antiferromagnetic ordering temperature $T_N = 125$ K is indicated.

Figure 2 | STM microscopy of Pr$\text{Sr}_2$Mn$_2$O$_7$. (a) 500×500 nm$^2$ micrograph of Pr$\text{Sr}_2$Mn$_2$O$_7$ collected at 78 K, 100 pA tunnel current and +0.8 V sample bias. The observed step height of 1.01±0.02 nm is consistent with $c/2 = 0.996$ nm (ref. 15). (b) 4×4 nm$^2$ micrograph (125 K, 50 pA, ~0.5 V) revealing a weak atomic lattice contrast, superposed on a 3–4 nm length scale inhomogeneity. A section through the micrograph is shown indicating the ~4 pm peak-to-peak corrugation of the atomic lattice. (c) 4×4 nm$^2$ micrograph (78 K, 100 pA, ~0.8 V) showing adatoms, vacancies and atomic rows; a section through an adatom is shown.
are similar to the surface inhomogeneities observed on the La$_{1.2}$Sr$_{1.2}$MnO$_4$ surface by STM$^{21,22}$ and surface X-ray scattering$^{23}$.

For the isostructural compound La$_{1.10}$Sr$_{1.10}$MnO$_4$, the surface exposed on cleaving has been identified as a (La,Sr)O plane on the basis of a combination of crystal structure arguments, STM imaging and X-ray photoelectron spectroscopy$^{24}$. In our STM images of PrSr$_2$MnO$_4$, all terrace step heights are integer multiples of c/2; no variation in step height is observed. Two possible mirror planes may yield a step height of c/2, bisecting either a single perovskite bilayer or a rock-salt layer. The bilayer is equivalent in structure to a perovskite manganite; these are known not to cleave naturally$^{16}$. The cleaving plane through the rock-salt layer is energetically favoured$^{25}$, implying that the observed surface is a (Pr,Sr)O layer.

**Imaging of adatoms and vacancies at the PrSr$_2$MnO$_4$ surface.**

Certain areas of the PrSr$_2$MnO$_4$ surface show a much increased roughness of around 250 pm peak-to-peak. Figure 2c shows a high-resolution STM topograph of such an area whereas Figure 3b shows a larger area image. The increased roughness is derived from a distribution of raised ‘adatoms’ and depressed ‘vacancies’. The adatom coverage is variable, with a maximum around 1% 65 surface unit cells. We may rule out the possibility that this rougher surface represents a different cleaving plane to the atomically flat surface, as terrace step heights are still restricted to multiples of c/2. Terrace steps of distinct heights, as observed on the single-layered manganite La$_{0.8}$Sr$_{1.2}$MnO$_3$, (ref. 22), indicating different surface terminations$^{26}$, were not seen for PrSr$_2$MnO$_4$.

On areas of the PrSr$_2$MnO$_4$ surface showing adatoms and vacancies in STM images, it is more difficult to achieve good atomic resolution than on the atomically flat surface. Nevertheless, atomic rows are visible in Figure 2c. Fourier transforms of larger scan areas (Fig. 3c) show peaks corresponding to the same lattice as the atomically flat surface (Fig. 3d), with measured lattice constant $a = 0.382 \pm 0.003$ nm. The same lattice is thus observed on both atomically flat and adatom-vacancy areas. No consistent ordering of adatoms was observed using either Fourier or self-correlation image analysis; the adatoms appear randomly distributed on the surface, with no reference to the atomic lattice.

The surface atomic lattice can also be observed by means of low energy electron diffraction (LEED). Figure 3e shows the LEED image of the same cleaved surface as in Figure 3b; the sample orientation is preserved on translation from the STM chamber to the LEED/sample preparation chamber while maintaining ultra-high vacuum. The LEED peaks correspond to a square lattice with $a = 0.43 \pm 0.03$ nm, indistinguishable from the STM value. LEED measurements have been recorded at a range of temperatures from 20 to 300 K, but no ‘satellite peaks’, indicative of surface charge ordering, appeared at any temperature.

**Scanning tunnelling spectroscopy of adatoms and vacancies.** We mapped spatial variations of the electronic density of states using scanning tunnelling spectroscopy (STS). Figure 4a shows an STM topograph of an area of the PrSr$_2$MnO$_4$ surface showing adatoms and vacancies. Figure 4b shows a conductance map, which has been derived from an STS map acquired simultaneously with the topograph. The conductance is displayed in filled states at a sample bias voltage of $-0.77$ V. There is a clear correlation between adatoms visible in the topographic image and peaks in the filled-state conductance. Figure 4c shows example spectra of adatoms, vacancies and the normal PrSr$_2$MnO$_4$ surface. Adatoms are associated with the normal PrSr$_2$MnO$_4$ surface.
an excess of filled-state tunnel current (representing integrated density of states) over the normal surface, and vacancies with a slight deficiency.

Figure 5a shows a topographic section that includes three vacancies and an adatom, and sections through two I(V) maps, at −0.8 V (filled states) and +0.8 V (empty states). The magnitude of the tunnel current, which represents the integrated density of states, is plotted along the same x axis used for the topographic section. The filled-state current shows a large peak at the position of the adatom, whereas the empty state current shows peaks at the positions of the vacancies. Figure 5b shows example I(V) spectra from adatoms, vacancies and the normal surface, from both I(V) maps. It is clear that the adatom spectra have a large excess of tunnel current at negative bias (filled states) compared with the normal surface, and that the tunnel current from vacancies is suppressed, whereas for positive bias (empty states) the reverse is true. On the basis of these data and the expectation that the surface is a (Pr/Sr)O layer, we interpret the bright adatoms seen in topographic images as oxygen, and the dark depressions as oxygen vacancies. This explains the large excess in filled states seen at adatom sites, as the oxygen will be negatively charged. Analogously, oxygen vacancies will carry a positive charge. A similar effect was seen in the I(V) spectra of La$_{0.6}$Ca$_{0.4}$MnO$_3$, in which an incomplete oxygen overlayer was formed on the MnO$_2$ terminated surface by annealing in an oxygen atmosphere.

As adatoms and vacancies are observed when the sample has been freshly cleaved, and the density of adatoms is not seen to increase even after 2 weeks in the ultra-high vacuum chamber, we suggest that the oxygen adatoms and vacancies appear not as a result of deposition or contamination, but that they arise when the sample is cleaved. Some oxygen ions may be removed with the cleaved-off part of the sample, forming surface vacancies. Similarly, oxygen ions from the cleaved-off part may be left on the surface as adatoms. Figure 5a represents the proposed model together with the topographic section; the [100] plane is shown, with Pr/Sr sites in yellow and O in red. The model is plausible as adatoms and vacancies occur in roughly equal numbers, as expected by symmetry. Contrast reversal is not observed between empty and filled-state images (see Supplementary Fig. S2), indicating that the ‘adatoms’ and ‘vacancies’ are not purely electronic contrast features but have physical height and depth.

**Oxygen adatom motion and adatom-vacancy recombination.** The oxygen adatom positions are not all fixed. Figure 6 (and Supplementary Movie 1) shows a time series of STM images in which several adatoms on the Pr$_3$Sr$_2$Mn$_2$O$_7$ surface appear or disappear between frames, although most adatoms remain stationary. We interpret this as hopping of adatoms from site to site, rather than single-site switching as the appearance and disappearance of adatoms are correlated. Indeed, we sometimes observe cascades in which adatoms appear to sequentially hop from one site to another. In Figure 6, it may be observed that the adatom site vacated at 520 s is reoccupied at 780 s, and then vacated again at 910 s. This effect has been observed multiple times and implies that adatoms have preferred positions on the Pr$_3$Sr$_2$Mn$_2$O$_7$ surface. As adatoms are not observed to form ordered arrays on the surface, these preferred positions are not determined by the atomic lattice. It is more likely that the surface inhomogeneities seen in Figures 2a and 3a, in which turn are likely due to charge inhomogeneities (mixed Mn valence and/or distribution of Pr$^{3+}$ and Sr$^{2+}$ ions), describe a potential landscape on which oxygen adatoms move. The observed mobility provides further evidence that these features are surface adatoms. Adatom mobility is strongly dependent on sample bias polarity, with negligible mobility at positive sample bias, suggesting an electrostatic component to the hopping mechanism. Mobile adatoms have been observed at a range of temperatures, with the adatom mobility increasing with sample temperature (see Supplementary Fig. S3, and Supplementary Discussion). The total number of adatoms and vacancies per image
decreases with increasing sample temperature (Supplementary Fig. S3). We attribute this to recombination of oxygen adatoms and vacancies, a process which we have observed directly. For example, Figure 7 shows two successive frames of a time series. In (a) an adatom and a vacancy are visible in close proximity whereas in (b) both the adatom and the vacancy have vanished, suggesting that the adatom has filled the vacancy. This can most clearly be seen in the comparative section (Fig. 7c). The observation of oxygen adatom motion provides a way to directly measure the surface oxygen exchange coefficient, a crucial parameter for the oxygen reduction action of solid oxide fuel cell cathodes\(^1,2\). Similarly, our observation of adatom-vacancy recombination demonstrates how adsorbed oxygen atoms may be incorporated into the cathode material.

**Bistable oxygen adatoms.** In addition to adatom hopping observed in time series of images, bistable adatoms appear in single (rastered) STM images and I(V) spectra. Such adatoms can switch between two adjacent positions several times during a single topographic scan. Figure 6d is an STM topograph including a bistable adatom, which moves laterally on the surface during the scan, resulting in a 'streaky' image of the adatom. Figure 7e shows two I(V) spectra collected at the same bistable adatom: these show the characteristic discontinuity caused by motion of the adatom. The spectra can be seen to switch between that typical of an adatom and that of the normal surface. The voltage is swept from positive to negative.

**Figure 6 | Adatom motion.** Time series of filled-state STM micrographs, 10×10 nm\(^2\), collected at 170 K (−0.8 V, 100 pA). The time per frame is 130 s and the total time elapsed is 910 s. Some adatoms (circled in adjacent frames) move from one frame to the next. See also Supplementary Movie 1.

**Figure 7 | Recombination of adatoms and vacancies.** (a, b) Successive frames in a time series showing recombination of an adatom/vacancy pair. Frames are 8×8 nm\(^2\) and are separated by 1 min. Data collected at 170 K (100 pA, +0.8 V). (c) Line profiles through a and b showing that both the adatom and the adjacent vacancy vanish. (d) 6×6 nm\(^2\) STM topograph showing a stable adatom (1) and a bistable adatom (3). The adatom appears streaky because it moves laterally between one scan line and the next; the STM tip takes around 10 s to scan over the adatom. Image collected at 125 K (100 pA, −0.8). (e) I(V) spectra at these positions, with a spectrum from the normal surface (position 2). Two spectra are shown from the bistable adatom: these show the characteristic discontinuity caused by motion of the adatom. The spectra can be seen to switch between that typical of an adatom and that of the normal surface.
Discussion

We have pioneered the visualization of oxygen defects and their motion for a perovskite oxide of manganese. Our discovery, including the choice of sample and low temperature cooling, enables future microscopic investigations of manganite fuel cell cathode materials. In such investigations, molecular oxygen would be bled into the STM vacuum chamber and the oxygen incorporation process at the manganite surface image, allowing the role of vacancies in oxygen reduction to be observed directly. Our work also resolves the long-standing puzzle of why the stable, atomically flat surfaces needed for STM are much less common for layered manganites than for cuprates, by showing clearly the surface roughness associated with oxygen defects at a manganite surface. Oxygen adatoms are mobile at temperatures well below room temperature, creating a highly challenging surface for STM investigations.

Oxygen motion has been held responsible for resistance switching in numerous studies of manganites \(^{8,10,12}\). Most studies indicate that electroresistance in manganite-based devices occurs at the interface between the manganite and the metal contact rather than in the bulk material \(^{8,10,12}\). The oxidation and reduction of either the manganite itself, or a metal oxide layer \(^{8,10,12}\), at the manganite–metal interface, as a result of electrochemical migration of oxygen ions, has been proposed as the underlying mechanism. Our work brings the benefit of atomic resolution to the study of electroresistance in manganites. Voltage-dependent oxygen motion and adatom bistability are observed at the interface between the manganite surface and the metal STM tip. We have demonstrated the unprecedented ability, not provided by mesoscopic and macroscopic devices, to view the voltage-induced displacement of single oxygen atoms at a manganite surface.

Methods

Sample growth and preparation. We used standard optical float zone methods to grow rod-like crystals of PrSr\(_{2}\)Mn\(_{2}\)O\(_{6}\) with typical diameter 5 mm and length 100 mm. PrSr\(_{2}\)Mn\(_{2}\)O\(_{6}\) crystals were cleaved in situ in ultra-high vacuum at 20 K and then immediately loaded into the STM, at 78 K. Cleaving at low temperature allowed reliable preparation of atomically flat and clean surfaces. Six cleaves were performed successfully on crystals from two batches.

STM experiments. STM measurements were recorded in ultra-high vacuum (\(<5 \times 10^{-11}\) mbar) using electrochemically etched tungsten tips derived from 250µm diameter wire. The experiments were conducted in the temperature range of 78–195 K. Images and I(V) spectra were typically collected at a tunnel current setpoint of 100 pA and a bias voltage of ±0.8 V. The STM piezo scanner was calibrated in three dimensions using topographic images of the Si[100] surface as a reference. Atomically flat PrSr\(_{2}\)Mn\(_{2}\)O\(_{6}\) surfaces have been observed at 78 and 125 K, whereas surfaces with adatoms and vacancies have been observed at a range of temperatures from 78 to 195 K. The two types of surface have been observed on the same sample in different locations. Both types of surface have been observed immediately after cleaving the PrSr\(_{2}\)Mn\(_{2}\)O\(_{6}\) crystal.

Dual mode STS mapping. For the data shown in Figure 5, dual mode STS mapping has been used to allow two STS measurements with different parameters to be collected simultaneously. The filled-state section is derived from an STS map in which the bias is swept from negative to positive, and the current is set at negative bias; the empty state section is derived from an STS map in which the bias is swept from positive to negative, and the current is set at positive bias. In both cases the current setpoint was 100 pA.

References

11. Tokunaga, Y. et al. Colossal electroresistance effect at metal electrode/ La\(_{0.6}\)Sr\(_{0.4}\)MnO\(_{3}\) interfaces. Appl. Phys. Lett. 88, 223507 (2006).
18. Fuchigami, K. et al. Tunable metallicity of the La\(_{0.6}\)Ca\(_{0.4}\)MnO\(_{3}\)(001) surface by an oxygen overlay. Phys. Rev. Lett. 102, 066104 (2009).
23. Wakabayashi, Y. et al. Surface effects on the orbital order in the single-layered manganite La\(_{0.6}\)Sr\(_{0.4}\)MnO\(_{3}\). Nat. Mater. 6, 972–976 (2007).

Acknowledgments

We thank K. Iwaya for guidance with STM methods, A.M. Stoneham and A. Fisher for helpful discussions, and C.F. Hirjibehedin, S. Wirth and H.M. Rønnow for assistance in preparing the paper. The work was supported by the Engineering and Physical Sciences Research Council of the United Kingdom and the European Union’s Framework Six programme.

Author contributions

Y. Tokunaga and Y. Tokura synthesized the samples; C.R., B.B. and G.A. designed the STM experiment; B.B. carried out STM measurements and bulk transport measurements; B.B., C.R. and G.A. provided interpretation of experimental results; B.B. and G.A. wrote the paper.

Additional information

Supplementary Information accompanies this paper at http://www.nature.com/naturecommunications

Competing financial interests The authors declare no competing financial interests.

Reprints and permission information is available online at http://npg.nature.com/reprintsandpermissions/

How to cite this article: Bryant, B. et al. Imaging oxygen defects and their motion at a manganite surface. Nat. Commun. 2:212 doi: 10.1038/ncomms1219 (2011).